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Activation of CuAlNi SMAs using a solar energy

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Abstract. Shape memory alloys are special materials that can perform mechanical work during martensitic transformation. CuAlNi shape memory alloys have gained attention based on them high transformation temperatures domain and can be considered as HTSMAs (high temperature shape memory alloys). The thermal component in order to achieve martensitic transformation can be obtained from sun using a solar concentrator. The heating/cooling process was registered and analyzed during the experiments. Material surface was analyzed before and after thermal shocks by microstructure point of view using scanning electron microscopy (SEM VegaTescan LMH II, SE detector) and atomic force microscopy (AFM EasyScan II, non-contact mode). The experiments follow the material behavior during fast heating and propose the possibility of activating smart materials using the sun heat for aerospace applications.

Keywords: shape memory alloy, solar energie, SEM, AFM, martensite.

1. Introduction

Cu-Al-Ni type alloys have been developed, due to the martensite thermoelasticity, in the commercial forms Cu-Al-Ni-X or CuAl-Ni-Mn-X, where X is an alloying element, with the role of finishing the structure which may be missing for certain Al/Ni reports [1, 2]. It can be seen that the values of Ms (martensite start) are very

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high, the (hyper) eutectoid alloys having interest having transformation points between 100-400 ºC with interest for high temperature applications (HTSMs) [3-6]. In addition, precipitation of the extremely hard γ2 phase cannot be suppressed even by a fast cooling stage. In order to eliminate the above disadvantages, the alliance with Ni was used. Following the introduction of nickel, the eutectoid moves to approx. 14% Al, concentration corresponding to a critical temperature Ms in the vicinity of the ambient temperature if the application requires [7]. If more than 5% Ni is added, γ2 precipitation is suppressed but NiAl precipitates may occur which are almost as fragile. For these reasons, the usual concentration of alloys with the shape memory based on Cu-Al-Ni is Cu- (10-14)% Al- (2-4)% Ni [8]. The typical structure of these alloys may contain: equilibrium phases: a solid solution α (cfc), isomorphic with copper, with the lattice parameter increasing with Al and Ni concentrations (a = 0.3658 nm for Cu-13% Al-4% Ni) [8, 9]; monoclinic NiAl intermetallic compound, with 3R structure and lattice parameters: a = 0.418 nm, b = 0.271 nm, c = 0.628 nm and β = 800; solid solution γ2, based on the electronic compound Cu9Al4, with complex cubic crystalline lattice with 52 atoms per elementary cell and parameter a = 868 nm [8]. The metastable phases: austenite β1, not transformed, based on the electronic intermetallic compound $CuAl_3$, with structure $D0_3$ and the network parameter a = 0.5836 nm [9, 10]; tension-induced monoclinic martensite, with 18R2 packing order and lattice parameters a = 0.443 nm, b = 0.533 nm, c = 3.819 nm and β = 890 [9]; thermally induced orthorhombic martensite, with the 2H packing order and the network parameters a = 0.439 nm, b = 0.5342 nm, c = 0.4224 nm; tension-induced monoclinic martensite α_1, with 6R packing order and lattice parameters a = 0.4503 nm, b = 0.5239 nm, c = 1.277 nm and β = 89.30.
In this article the activation of CuAlNi shape memory alloy using solar energy is analyzed by means of behavior of the alloy at 10 thermal shocks cycles till 300 respectively 600 ºC with high heating/cooling rates.

2. Experimental details

The heating experiments were realized on a solar concentrator, part of the Promes - France laboratories. PROMES - CNRS at Font-Romeu Odeillo has a whole range of systems based on solar flux that can be used as experimental furnaces or heating systems, figure 1. Intelligent materials like SMAs (shape memory alloys) were activated with help from solar beam reflected by a mechanized metallic window (5 x 12m) situated at the bottom of the building, through a shutter system (used to control the intensity of the light beam) and a concave concentrator, as presented schematically in figure 1. The beam is transmitted at a focus point 1-1.5 m down to a center.
In order to locate the sample in the center of light concentration and to benefit of the biggest intensity and temperature we use an aluminum trolley, top figure 1) that can be translated on X-Y axis [4, 11]. The aluminum sustaining system is continuously cooled with water in order to avoid accidents. The metallic experimental probes,

squares of 10x10 mm with a K thermocouple, used to retrieve data about the temperature evolution in time, connected to a Graphtec Corporation equipment type GL220 [3, 12]. Both heating and cooling stages were registered and analyzed. The thermal shocks were performed moving the trolley from and under the sun light [13-15]. The heating experimental temperatures were 300 and 600°C ± (10 respectively 25 because of the thermal inertia of the metallic samples).

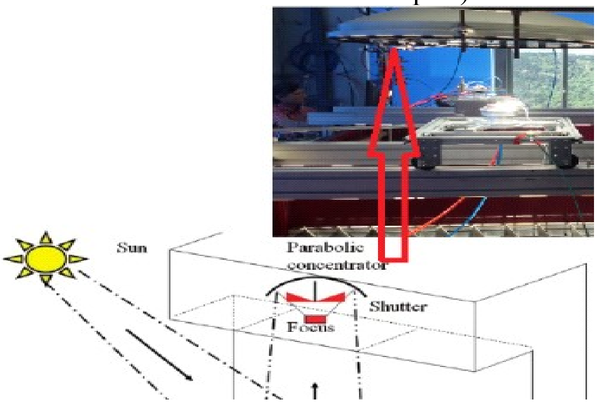
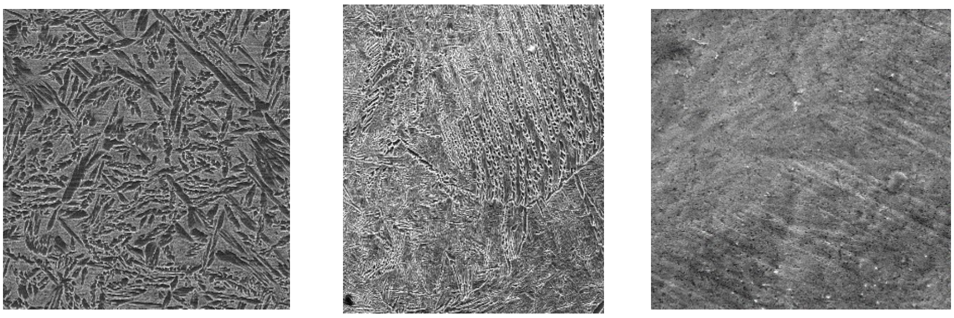


Fig. 1. Solar furnace set-up for heating up CuAlNi alloy.

Copper based shape memory alloy (Cu12Al4Ni) as 10x10 mm square were used for exposure to heat (10 cycles) at 300 and 600°C. The flux was applied in the middle part of the sample by situating the sample in the middle of the solar flux. The temperature increasing and decreasing was registered using a K type thermocouple and the heat was analyzed from the middle of the sample. The heating/cooling shape of curves between time and temperature variation for all 10 curves are already presented and discussed in [3, 4]. The heating rates are between 10 to 80 ºC/ms and the cooling rates between 12 and 70 ºC/ms. Using a shutters system, we can control the intensity of the light beam and a trolley for positioning the sample in the middle of the fascicule.

3. Experimental results

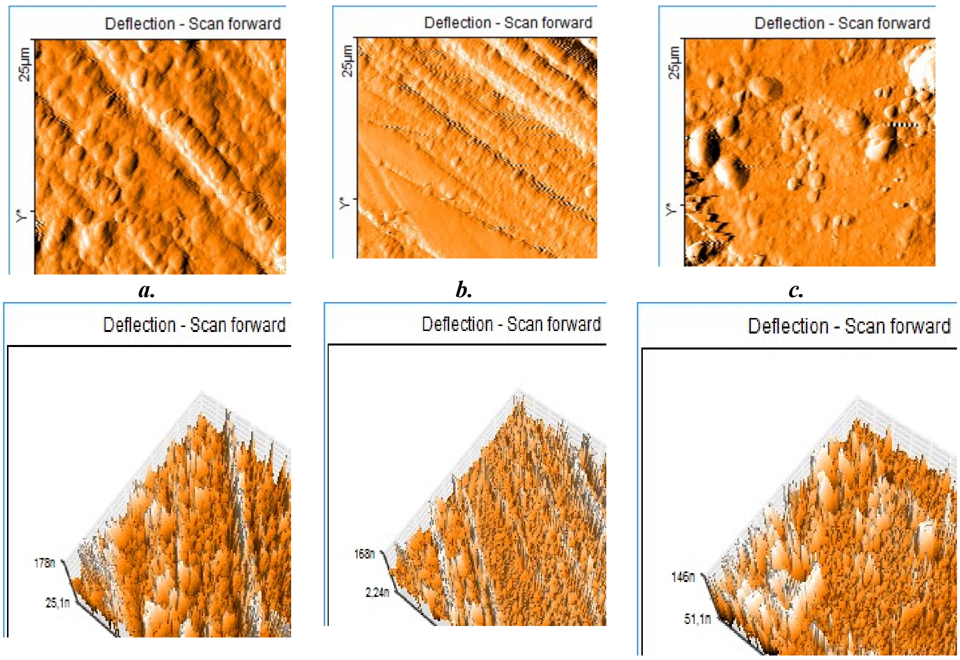
After the 10 heating/cooling cycles at 300 or 600 ºC the surface of the experimental samples was investigated through SEM and AFM techniques. In figure 2, SEM images of CuAlNi alloy are presented as a) initial state, b) after 10 heating/cooling cycles to 300 ºC and c) after 10 heating/cooling cycles to 600 ºC.
Phase β, based on the Cu_3Al intermetallic compound, represents austenite which, at very slow cooling, decays eutectoid at 570ºC [16-19] resulting in a solid isomorphic solution with copper (α, cfc) and a solid solution based on the intermetallic compound of Cu_9Al_4 type (γ_2, complex cube with 52 atoms per elementary cell), figure 2 c) [20]. When cooling at normal speeds, the austenite β (A_2) is ordered becoming β1 ($D0_3$), at approx. 525 ºC. The same thing happens with the solid solution α which is ordered at short distances and becomes α_2 [21, 22].



a. b. c.

Fig. 2. SEM images of CuAlNi alloy: a) initial state, b) after 10 heating/cooling cycles to 300 ºC and c) after 10 heating/cooling cycles to 600 ºC.

The microstructure profile at the surface was analyzed, at a smaller scale, using AFM technique. In figure 3 are presented AFM images of CuAlNi surface a), d) initial state, b), e) after 10 cycles at 300 ºC and c), f) after 10 cycles at 600 ºC. The results don't present modifications in case of thermal solicitation till 300 ºC (10 cycles), figure 3 comparison between a) and b), in 2D and d) and e) for 3D images. In case of heating to 600 ºC a modification of the microstructure relief can be observed, comparison between a) and c), confirming the observations from SEM analyze, figure 2 a) and c).



d. e. f.

Fig. 3. AFM images of CuAlNi surface a), d) initial state, b), e) after 10 cycles at 300 ºC and c), f) after 10 cycles at 600 ºC.

As the heat-induced martensite plates, from Cu-Al-Ni alloys, continuously increase upon cooling and continuously shorten upon heating [23] can conclude that "there is a balance or exchange between chemical free energy and free energy. elastic deformation at transformation " [22]. This martensitic transformation, with thermal hysteresis of the order of 30K [25], corresponds to a simple memory effect of the shape of approx. 4% and a two-way shape memory effect of approx. 2% [24]. Thermal cycling of the Cu-Al-Ni MFA produces substantial material stiffening, especially when retention cycles are applied to traction (heating-cooling under sustained elongation) [26]. Normally, thermal cycling leads to an increase in the density of dislocations, which causes the displacement of critical temperatures. This phenomenon will have to be further investigated by DSC (differential calorimetry) analysis.

4. Conclusions

After we analyze the experimental results we can present few conclusions:

- Thermal activation of metallic elements made of CuAlNi – SMAs is easily done by sun energy,
- Different temperatures can be reached with different heating rates based on the application needs,
- Solar concentrators (with different dimensions) can be considered a cheap, inexpensive and proper solution for shape memory alloy functionalization in inaccessible spaces or for aerospace applications,
- Heating the surface of CuAlNi till 300 and 600 ˚C produce a modification of the chemical composition by forming oxides (copper based most of them) identified through SEM and AFM analyses,
- SEM and AFM results present no modification at the primary plates dimensions and orientation also after the thermal shocks no secondary plates formation was observed in case of heating the material till 300 ºC but when we analyze the sample heated to 600 ºC no more martensite plates appear based on a high heating stage and also a relatively high cooling rate, further investigation on the relief modification must be realized by means of the depth of the modification,
- New experiments with thermal shocks are necessary in a vacuum state in order to analyze the material behavior in aerospace conditions.

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